

***In-situ* and *Ex-situ* Ohmic Contacts To Heavily Doped p-InGaAs**

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Very low resistance metal-semiconductor contacts are crucial for the performance for THz-bandwidth transistors. $\rho_c < 1 \times 10^{-8} \Omega\text{-cm}^2$ is required for III-V HBTs and FETs for having simultaneous 1.5 THz f_i and f_{max} [1, 2]. We had earlier reported low resistivity *in-situ* ($\rho_c = 1.1 \times 10^{-8} \Omega\text{-cm}^2$) [3] and *ex-situ* ($\rho_c = 1.4 \times 10^{-8} \Omega\text{-cm}^2$) [4] ohmic contacts to n-In_{0.53}Ga_{0.47}As; contacts of similar resistivity to p-In_{0.53}Ga_{0.47}As are required for several InP-based devices, most particularly for the base contacts of HBTs [5, 6]. Low resistivity is less readily obtained for *ex-situ* contacts than it is for *in-situ* contacts, as semiconductor surface oxides must be removed prior to metal contact formation. While some device fabrication processes allow *in-situ* p-semiconductor-metal contact formation, common HBT fabrication processes force base contact formation after the semiconductor surface is exposed to air, hence low-resistivity *ex-situ* contacts are required. *Ex-situ* Pd/Ti/Pd/Au contacts have shown $\rho_c = 4 \times 10^{-8} \Omega\text{-cm}^2$ to p-In_{0.53}Ga_{0.47}As [6, 7]. Here we report record low contact resistivities of $\rho_c = (1.0 \pm 0.7) \times 10^{-8} \Omega\text{-cm}^2$ and $(1.5 \pm 0.9) \times 10^{-8} \Omega\text{-cm}^2$ to p-In_{0.53}Ga_{0.47}As obtained by *in-situ* and *ex-situ* techniques, respectively, using iridium (Ir) contact metal.

The semiconductor epilayers were grown by solid source MBE. A 100 nm undoped In_{0.52}Al_{0.48}As layer was grown on a semi-insulating InP (100) substrate, followed by 100 nm of carbon doped In_{0.53}Ga_{0.47}As, using CBr₄ as the dopant source. The In_{0.53}Ga_{0.47}As layer was grown at 350 °C substrate temperature and a 12:1 groupV to groupIII ratio. Ir (20 nm) and Mo (20 nm) were deposited *in-situ* on half the wafer surface in an electron beam evaporator attached to the MBE chamber under ultra high vacuum (UHV). The wafer was removed from vacuum and cleaved; the piece having *in-situ* Ir/Mo metal is the *in-situ* sample; the piece having a bare In_{0.53}Ga_{0.47}As surface is the *ex-situ* sample.

The *ex-situ* sample was exposed to UV-ozone for 30 minutes and then treated with 1:10 HCl:H₂O and a DI rinse for 1 minute each. It was then loaded in the MBE chamber, cleaned with atomic hydrogen at 65 °C for 30 minutes, and annealed at 450 °C for 10 minutes in a separate chamber connected under UHV. After this anneal, the RHEED pattern showed a clear (3×2) reconstruction. Ir (20 nm) and Mo (20 nm) were then deposited in the electron beam evaporator connected under UHV.

Both *in-situ* and *ex-situ* samples were processed into transmission line model (TLM) structures for contact resistance measurement. Ti (20 nm)/Au (500 nm)/Ni (50 nm) contact pads were patterned on the samples using photolithography and lift-off after an e-beam deposition (Fig. 1). Ir/Mo was then dry etched in SF₆/Ar plasma using Ni as a mask. Resistance was measured by four-point (Kelvin) probing using separate pads for current biasing and voltage measurement (Fig. 2). The processed samples were annealed under nitrogen atmosphere at 250 °C for 60 minutes, replicating the thermal cycle experienced by a base contact during heterojunction bipolar transistor fabrication.

As determined through Hall measurements, the hole concentration, mobility and sheet resistance on the as-grown In_{0.53}Ga_{0.47}As sample were $1.5 \times 10^{20} \text{ cm}^{-3}$, 27.1 cm²/Vs and 164.8 Ω/□, respectively. Figure 3 shows the variation of measured TLM resistance with gap spacing for the *in-situ* and *ex-situ* samples. The ρ_c achieved for the un-annealed *in-situ* and *ex-situ* samples were $(1.0 \pm 0.6) \times 10^{-8} \Omega\text{-cm}^2$ and $(1.5 \pm 0.9) \times 10^{-8} \Omega\text{-cm}^2$, respectively, these being the lowest reported to date for contacts to p-type In_{0.53}Ga_{0.47}As. Contact resistivities of the *in-situ* and *ex-situ* annealed samples were $(1.2 \pm 0.7) \times 10^{-8} \Omega\text{-cm}^2$ and $(1.8 \pm 0.9) \times 10^{-8} \Omega\text{-cm}^2$, respectively, differing from the pre-anneal data by less than the precision in measurement. These ultra low contact resistivities make Ir a strong candidate for p-type contacts in THz HBTs.

50 nm Ni
500 nm Au
20 nm Ti
20 nm Mo
20 nm Ir
100 nm $\text{In}_{0.53}\text{Al}_{0.47}\text{As}$: C (p-type)
100 nm $\text{In}_{0.52}\text{Al}_{0.48}\text{As}$: NID buffer
Semi-insulating InP: Substrate

Fig 1: Cross-section schematic of the metal-semiconductor contact layer structure. Ir and Mo were deposited in an electron beam deposition chamber connected to MBE system under ultra high vacuum.

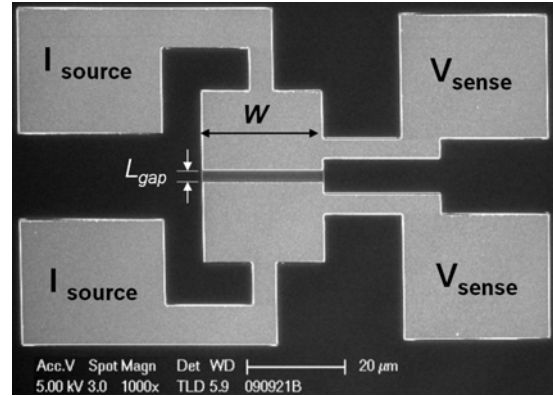


Fig 2: Scanning electron micrograph of the TLM pattern used for the contact resistivity (ρ_c) measurement. Separate pads were used for current biasing and voltage measurement.

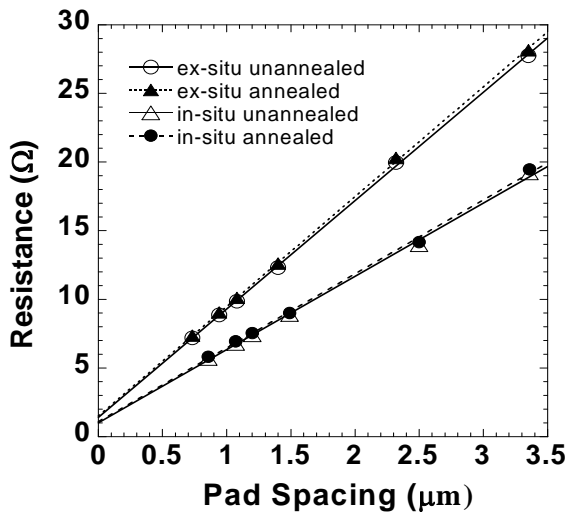


Fig 3: Measured TLM resistance as a function of pad spacing for un-annealed and annealed *in-situ* and *ex-situ* Ir contacts on p- $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$.

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