

***In-situ* Iridium Refractory Ohmic Contacts to p-InGaAs**

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Very low resistance metal-semiconductor contacts are crucial for the performance of transistors in THz bandwidths. The base and emitter contact resistivities (ρ_c) in heterojunction bipolar transistors (HBTs) must decrease in proportion to the inverse square of the transistor cutoff frequency [1, 2]. A ρ_c of less than $1 \times 10^{-8} \Omega\text{-cm}^2$ is required for III-V HBTs and FETs for having simultaneous 1.5 THz f_t and f_{max} [1, 2]. Ohmic contacts to p-type $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$ have been studied extensively because of its application as the base contacts in InP based HBTs. *Ex-situ* Pd/Ti/Pd/Au contacts have shown $\rho_c = 4 \times 10^{-8} \Omega\text{-cm}^2$ to p- $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$ [3, 4], but penetrates into the semiconductor by combined chemical reaction and diffusion [5]. Low resistivity, thermally stable contacts having < 5 nm metal penetration depth are required for HBTs having < 20 nm thick base layers. Here we report $\rho_c = (5.8 \pm 4.8) \times 10^{-9} \Omega\text{-cm}^2$ for in-situ, refractory Ir contacts to p-type $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$.

The semiconductor epilayers were grown by solid source MBE. A 100 nm undoped $\text{In}_{0.52}\text{Al}_{0.48}\text{As}$ layer was grown on a semi-insulating InP (100) substrate, followed by 100 nm of carbon doped $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$, using CBr_4 as the dopant source. The $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$ layer was grown at 350 °C substrate temperature and a 7:1 group V to group III ratio. Ir (20 nm) and Mo (20 nm) were deposited *in-situ* on half the wafer surface in an electron beam evaporator attached to the MBE chamber under ultra high vacuum (UHV). Hall measurements were done on the samples not coated with Ir/Mo. Samples coated with Ir/Mo were processed into transmission line model (TLM) structures for contact resistance measurement. Ti (20 nm)/Au (500 nm)/Ni (50 nm) contact pads were patterned on the samples using photolithography and lift-off after an e-beam deposition. Mo was then dry etched in an SF_6/Ar plasma using Ni as a mask. The TLM structures were then isolated using mesas formed by photolithography and a subsequent wet etching with 1:1:25: H_3PO_4 : H_2O_2 :DI water. Resistances were measured by four-point (Kelvin) probing. The processed samples were annealed under nitrogen atmosphere at 250 °C for 60 minutes, replicating the thermal cycle experienced by a base contact during transistor fabrication.

As determined through Hall measurements, the highest hole concentration (p) achieved was $2.2 \times 10^{20} \text{cm}^{-3}$ exhibiting a mobility (μ) and sheet resistance (R_{sh}) of 29.9 cm^2/Vs and 94 Ω , respectively. The ρ_c achieved for the sample with $2.2 \times 10^{20} \text{cm}^{-3}$ holes was $(5.8 \pm 4.8) \times 10^{-9} \Omega\text{-cm}^2$, which is the lowest reported to date for ohmic contacts to p-type InGaAs. The sheet resistivity determined from TLM measurements was 99 Ω , correlating closely to the sheet resistivity obtained with Hall measurements. The annealed samples show a ρ_c of $(8.0 \pm 5.6) \times 10^{-9} \Omega\text{-cm}^2$, differing from the un-annealed samples by less than the margin of error in measurement.

Given the low ρ_c and demonstrated thermal stability, in-situ Ir is a strong candidate for base ohmic contacts in THz HBTs.

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50 nm Ni
500 nm Au
20 nm Ti
20 nm Mo
20 nm Ir
100 nm $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}$: C (p-type)
100 nm $\text{In}_{0.52}\text{Al}_{0.48}\text{As}$: NID buffer
Semi-insulating InP: Substrate

Fig 1: Cross-section schematic of the metal-semiconductor contact layer structure. Ir/Mo were deposited in an electron beam deposition system connected to MBE system under ultra high vacuum.

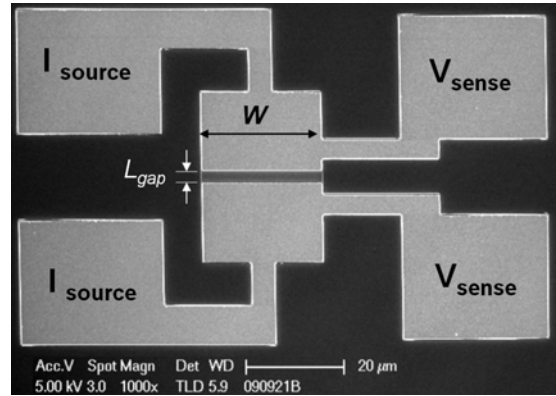


Fig 2: Scanning electron micrograph of the TLM pattern used for the contact resistivity (ρ_c) measurement. Separate pads were used for current biasing and voltage measurement.

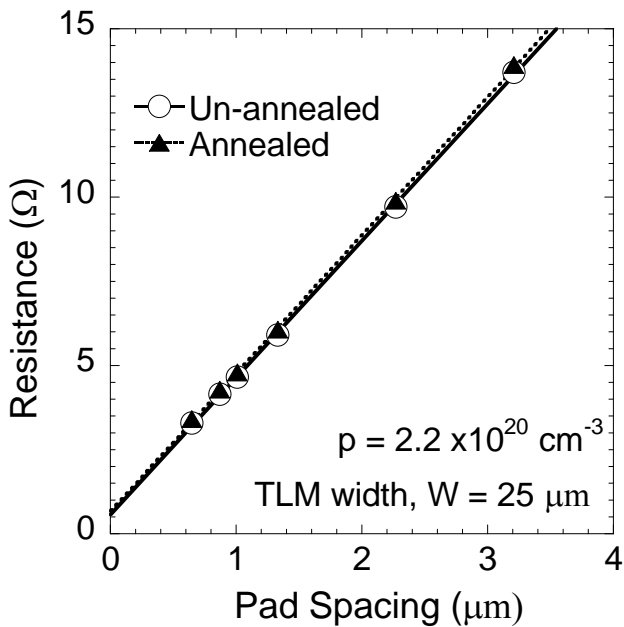


Fig 3: Measured TLM resistance as a function of pad spacing for un-annealed and annealed Ir contacts on p-InGaAs.

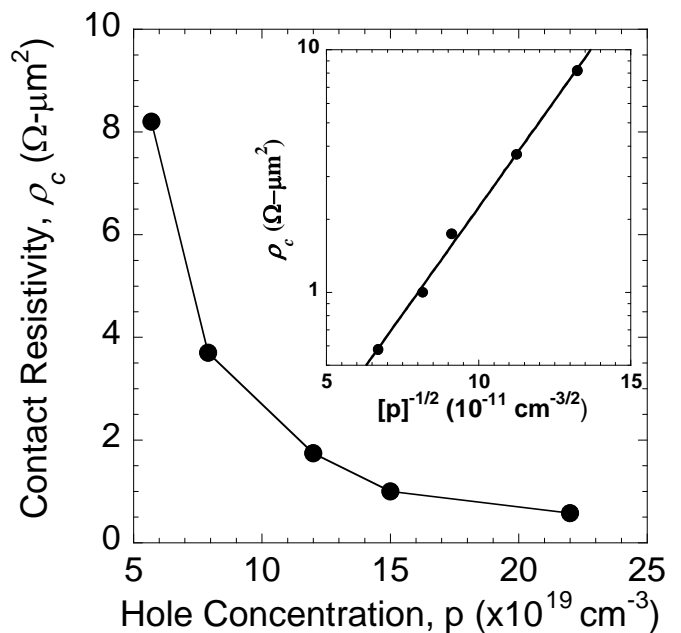


Fig. 4: Variation of contact resistivity (ρ_c) with hole concentration. Inset: Exponential dependence of ρ_c on $(p)^{-1/2}$ indicating tunneling behavior of the contacts.